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Synthesis and Properties of Poly(1-chloro-1-alkynes)

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ABSTRACT: 1-Chloro-1-hexyne, -1-octyne, and -1-decyne (ClC=CR; R = n-C₄H₉, n-C₆H₁₃, and n-C₈H₁₇) polymerized with Mo catalysts [e.g., $MoCl_5$, $MoCl_5$ —n- Bu_4Sn , and $Mo(CO)_6$ — CCl_4 — $h\nu$] in high yields to give new high molecular weight polymers (M_w ca. $2 \times 10^5 - 2 \times 10^6$). 1-Chloro-1-hexadecyne, which has a long alkyl group $(R = n-C_{14}H_{29})$, polymerized similarly, whereas 1-chloro-3-propyl-1-hexyne, which has a sec-alkyl group $[R = CH(n-C_3H_7)_2]$, hardly polymerized owing to the steric reason. The product polymers had the structure of -(CCl=CR)_n- according to the spectroscopic analyses. The polymers of 1-chloro-1-hexyne, -1-octyne, and -1-decyne were white solids, while poly(1-chloro-1-hexadecyne) was a white rubbery material. They were totally soluble in many organic solvents, thermally fairly stable in air, amorphous, film-forming, electrically insulating, and nonparamagnetic.

Introduction

Polymers obtained from substituted acetylenes have alternating double bonds in the main chain and various groups in the side chain. Unlike the unsubstituted polyacetylene, many of them are stable to air and soluble in organic solvents. Catalysts based on groups 5 and 6 transition metals (Nb, Ta, Mo, and W) have been found effective in their synthesis.2

Many 1-halo-1-alkynes have been prepared so far.³ The halogen atoms in 1-halo-1-alkynes are expected to have various effects such as inductive, mesomeric, and steric ones on the C=C bond, which has incited many organic chemists to study their reactivity. In general, 1-chloro-1-alkynes are more stable in air at high temperature than other 1-halo-1-alkynes and hence are easier to handle.

To date, several polymers of 1-halo-1-alkynes have been known: ClC=CPh, ClC=C-n-C₆H₁₃, BrC=CPh, and BrC≡C(CH₂)₄Br. 1-Chloro-2-phenylacetylene polymerizes with only Mo catalysts to give a totally soluble high molecular weight polymer (MW ~ 1 × 10⁶).^{5a,5b} A recent short communication by us has revealed that 1-chloro-1octyne also yields a high molecular weight polymer.5c In contrast, the polymer molecular weights of 1-bromo-1alkynes (BrC≡CPh^{5d} and BrC≡C(CH₂)₄Br^{5e}) are no more than 1×10^4 . Thus, it seems of great interest to examine the polymeriztion of various aliphatic 1-chloro-1-alkynes in detail.

The present paper deals with the synthesis and properties of poly(1-chloro-1-alkynes). New, high molecular weight polymers have been successfully synthesized from linear 1-chloro-1-alkynes (1-chloro-1-hexyne, -1-octyne, -1-decyne, and -1-hexadecyne) and their structure and properties have been clarified.

Results and Discussion

Polymerization of 1-Chloro-1-hexyne, 1-Chloro-1octyne, and 1-Chloro-1-decyne. Table I gives results on the polymerization of three alkynes with C₄-C₈ alkyl groups, especially 1-chloro-1-octyne, in the presence of various catalysts. 1-Chloro-1-octyne polymerized only with Mo catalysts and did not yield any polymer (methanolinsoluble product) with W, Nb, Ta, Fe, or Ti catalysts.

Difficulty of this monomer to polymerize with W catalysts coincides with the general tendency of 1-halo-1-alkynes,5 and it should be associated with the electron-withdrawing property of the chlorine atom. It is known that Nb and Ta catalysts work well even for the polymerization of sterically crowded acetylenes as 1-(trimethylsilyl)-1propyne.^{2a} Hence, the result that 1-chloro-1-octyne does not polymerize with Nb or Ta catalysts is attributable to the electronic effect of the monomer. There has been no example of disubstituted acetylenes polymerizable with Ziegler catalysts (e.g., Fe and Ti catalysts); the present result for 1-chloro-1-octyne is consonant with it.

1-Chloro-1-hexyne, -1-octyne, and -1-decyne polymerize even with MoCl₅ alone in high yields under the conditions shown in Table I. However, use of n-Bu₄Sn, which reduces and/or alkylates MoCl₅, as cocatalyst accelerates the polymerizations (see below) and enhances the $\bar{M}_{\rm w}$'s of polymers close to 1×10^6 . The catalyst obtained by UV irradiation of the CCl₄ solution of Mo(CO)₆ effects still higher $M_{\rm w}$'s around 2×10^6 . Thus, the Mo(CO)₆-CCl₄- $h\nu$ catalyst is suitable to accomplish very high molecular weight (the MoCl₅-cocatalyst systems, however, surpass it with regard to their higher activity and the ease of their preparation).

Table II shows what organometallics are useful as cocatalysts. As seen in the polymerization of 1-chloro-1octyne, organometallics containing groups 4 and 5 maingroup metals (Sn, Si, Sb, and Bi) generally give high yields of polymers with high molecular weights. In contrast, with organolithiums and -aluminum were formed appreciable amounts of methanol-soluble oligomers. This can be atttributed to their rather too strong reducing and alkylating ability (this is commensurate with the fact that MoCl₅ is excessively reduced to precipitate partly in the presence of these cocatalysts). Further, no polymer formed at all with the use of i-Bu₂AlH or 9-BBN, both of which have a metal-H bond.

For 1-chloro-1-hexyne, -1-octyne, and -1-decyne, groups 4 and 5 organometallic cocatalysts gave clearly better results than those in the absence of any cocatalyst; that is, both the yield and $\bar{M}_{\rm w}$ of polymers tended to increase (Table II). In the case of organosilicons, no cocatalyst effect was observed unless they had a Si-H bond (e.g.,

Table I Polymerization of 1-Chloro-1-hexyne, 1-Chloro-1-octyne, and 1-Chloro-1-decyne by Various Catalysts^a

				polymer ^b	
no.	catalyst	monomer convn, %	yield, %	$ar{M}_{ m w}/10^{3 m c}$	$ar{M}_{ m n}/10^{3c}$
		1-Chloro-1-hexy	ne		
1	$MoCl_5$	100	91	320	170
2	$MoCl_5-n$ -Bu $_4$ Sn	100	94	890	590
3	$Mo(\mathring{CO})_6$ - \mathring{CCl}_4 - hv^d	97	81	2600	850
		1-Chloro-1-octy	ne		
4	$MoCl_5$	100	78	220	96
5	$MoCl_5-n$ -Bu ₄ Sn	100	92	870	510
6	$Mo(CO)_6$ - CCl_4 - $h\nu^d$	90	85	1800°	740
7	WCl_6	10	0		
8	WCl ₆ − <i>n</i> -Bu ₄ Sn	16	0		
9	$W(CO)_6$ - CCl_4 - $h\nu^d$	9	0		
10	$NbCl_5-n-Bu_4Sn^f$	100g	0		
11	$TaCl_5-n$ - Bu_4Sn^f	62^g	0		
12	$Fe(acac)_3-3Et_3Al$	8	0		
13	$Ti(O-n-Bu)_4-4Et_3Al$	42^g	0		
	•	1-Chloro-1-decy	ne ·		
14	$MoCl_5$	100	72	310	180
15	$MoCl_5-n$ -Bu $_4$ Sn	100	96	940	550
16	$Mo(CO)_6$ - CCl_4 - $h\nu^d$	96	92	1600	790

^a Polymerized in toluene at 30 °C for 24 h; $[M]_0 = 0.50$ M, $[Cat] = [n\text{-Bu}_4\text{Sn}] = 20$ mM. ^b Polymer: methanol-insoluble product. ^c Determined by GPC. ^d Polymerized in CCl_4 ; $[M(CO)_6] = 10$ mM. ^e $[\eta] = 6.96$ dL/g (in toluene, 30 °C). ^f Polymerized at 80 °C. In the absence of n-Bu₄Sn, conversions were 100% (NbCl₅) and 84% (TaCl₅), and no polymer formed. Product: cyclotrimers and/or linear oligomers.

Table II Effects of Organometallic Cocatalysts on the Polymerization of 1-Chloro-1-hexyne, 1-Chloro-1-octyne, and 1-Chloro-1-decyne by MoCl₅^a

	1-Chioro-1-decyne by Mocis							
	monomer	polymer ^b						
cocatalyst	convn, %	yield, %	$ar{M}_{ m w}/10^{3c}$	$ar{M}_{ m n}/10^{3c}$	$[\eta], \mathrm{dL}/\mathrm{g}^d$			
	1-Chloro-1-hexyne							
none	100	91	320	170				
n -Bu $_4$ Sn	100	94	890	590				
$\mathrm{Et_{3}SiH}$	100	98	1300	540				
Ph_3Sb	100	67	870	580				
Ph₃Bi	100	98	700	300				
n-BuLi	100	65	870	480				
$\mathrm{Et}_{3}\mathrm{Al}$	96	16	910	540				
		1-Chloro	1-octyne					
none	100	78	220	96	0.95			
n-Bu ₄ Sn	100	92	870	510	2.80			
Ph₄Sn	100	82	640	210				
Et ₃ SiH	100	93	1200	570				
Ph_3SiH	100	88	510	180				
Ph_3Sb	100	84	650	230				
Ph_3Bi	100	93	520	230	2.16			
n-BuLi	100	48	820	390	3.78			
t-BuLi	90	45	830	380				
$\mathrm{Et_{3}Al}$	100	55	780	340	2.89			
i-Bu ₂ AlH	100	0						
9-BBN ^e	96	0						
		1-Chloro-	1-decyne					
none	100	72	310	180				
n-Bu₄Sn	100	96	940	550				
Et ₃ SiH	100	98	1400	560				
Ph_3Sb	100	82	720	360				
Ph_3Bi	100	98	560	330				
n-BuLi	27	18	870	440				
$\mathbf{Et_3Al}$	84	28	730	260				

^a Polymerized in toluene at 30 °C for 24 h; [M]₀ = 0.50 M, [Cat] = [Cocat] = 20 mM. bPolymer: methanol-insoluble product. ^c Determined by GPC. ^d Intrinsic viscosities measured in toluene at 30 °C. e9-BBN: 9-borabicyclo[3.3.1]nonane.

Et₄Si); this should be due to the Si-C bond being stronger than those of Sn-C, etc.

Polymerization of 1-chloro-1-octyne proceeded in hydrocarbons and halogenated hydrocarbons (Table III).

Table III Solvent Effects on the Polymerization of 1-Chloro-1-octyne by MoCl₅-n-Bu₄Sn^a

	monomer	polymer ^b			
solvent	convn, %	yield, %	$ar{M}_{ m w}/10^{3 m c}$	$M_{\rm n}/10^{3c}$	
toluene	100	92	870	510	
cyclohexane	100	92	560	160	
n-hexane	100	93	580	260	
PhCl	100	93	400	190	
CCl ₄	100	92	430	170	
PhOCH ₃	100	88	240	100	
PhCOOČH ₃	44	0			
PhCOCH ₃	0	0			

^a Polymerized at 30 °C for 24 h; $[M]_0 = 0.50 \text{ M}$, $[MoCl_5] = [n-1]$ $Bu_4Sn] = 20 \text{ mM}.$ ^b Polymer: methanol-insoluble product. Determined by GPC.

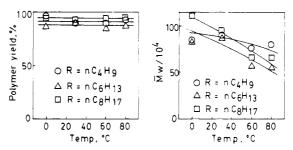


Figure 1. Temperature dependence of the polymerization of ClC=CR by MoCl₅-n-Bu₄Sn (1:1) (in toluene, 24 h, [M]₀ = 0.50 $M, [MoCl_5] = 20 \text{ mM}).$

The polymer yield was hardly dependent on the kind of these solvents. However, the $\bar{M}_{\rm w}$ changed to some extent with the kind of solvents, the highest value being attained in toluene. Regarding oxygen-containing solvents, 1chloro-1-octyne gave a polymer in anisole, but not in more polar solvents like methyl benzoate and acetophenone. Thus, solvents such as toluene that have low polarity and dissolve catalysts and polymers well are favorable for polymerization.

Effects of temperature on the polymerization of the three 1-chloro-1-alkynes by MoCl₅-n-Bu₄Sn were studied (Figure 1). In the temperature range 0-80 °C, the polymer

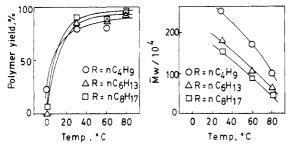


Figure 2. Temperature dependence of the polymerization of ClC=CR by $Mo(CO)_6$ -CCl₄- $h\nu$ (in CCl₄, 24 h, [M]₀ = 0.50 M, [Mo(CO)₆] = 10 mM).

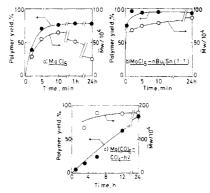


Figure 3. Time courses of the polymerization of 1-chloro-1-octyne by Mo catalysts (in toluene, 30 °C, $[M]_0 = 0.50 M$, $[MoCl_5] = 20 mM$, $[Mo(CO)_6] = 10 mM$).

yield was always as high as 90% with every monomer. On the other hand, the $\bar{M}_{\rm w}$ tended to decrease with increasing temperature. Therefore, 0 or 30 °C is suitable to obtain high molecular weight polymers.

In the case of the Mo(CO)₆–CCl₄– $h\nu$ catalyst, polymers were formed in ~90% yields at 30 °C or above (Figure 2). The polymer yields at 0 °C were low or negligible with this catalyst. The $\bar{M}_{\rm w}$ values at 30 °C were around 2 × 10⁶ and decreased sharply with increasing temperature. Hence, 30 °C is the optimal temperature with the Mo(CO)₆–CCl₄– $h\nu$ catalyst.

1-Chloro-1-octyne polymerization proved to take a different time course for the yield and $\bar{M}_{\rm w}$ of polymer depending on the kind of Mo catalysts (Figure 3): In the polymerization by MoCl₅ alone, all the monomer was consumed after 10 min, and the polymer yield at that time was 76%. The $\bar{M}_{\rm w}$ of the polymer after 10-min reaction was 6.5×10^5 , but it decreased to 2.2×10^5 after 24-h reaction owing to polymer degradation (\bar{M}_w/\bar{M}_n) was practically independent of polymerization time) (Figure 3a). With MoCl₅-n-Bu₄Sn as catalyst, the monomer consumption was completed within 2 min, and then the polymer yield was 96% and $\bar{M}_{\rm w}$ was 6.8×10^5 . No decrease of \bar{M}_{m} was observed even after 24-h polymerization (Figure 3b). The polymerization by $Mo(CO)_6$ - CCl_4 - $h\nu$ proceeded much more slowly, but the polymer yield reached 85% after 24 h. The polymer formed possessed high $\bar{M}_{\rm w}$ values $(1.4 \times 10^6 - 1.8 \times 10^6)$ irrespective of conversion (Figure 3c).

It is clear from Figures 1–3 that $Mo(CO)_6$ – CCl_4 – $h\nu$ is less active than $MoCl_5$ –n-Bu₄Sn but produces polymers having higher molecular weights.

Polymerization of 1-Chloro-1-hexadecyne (ClC=n- $C_{14}H_{29}$). 1-Chloro-1-hexadecyne was chosen as a monomer which has a very long linear alkyl group, and its polymerization was examined. As seen in Table IV, suitable MoCl₅-cocatalyst systems afforded polymers in high yields (80-90%), while Mo(CO)₆-CCl₄- $h\nu$ achieved a very high molecular weight ($\bar{M}_{\rm w}$ 1.8 × 10⁶). Thus, the

Table IV,

Polymerization of 1-Chloro-1-hexadecyne
(ClC=C-n-C₁₄H₂₉)^a

	catalyst	monomer convn, %	polymer ^b		
no.			yield, %	$ar{M}_{ m w}/10^{3c}$	$ar{M}_{ m n}/10^{3c}$
1	MoCl ₅	100	70	150	56
2	MoCl ₅ −n-Bu ₄ Sn	100	87	630	320
3	MoCl ₅ -Et ₃ SiH	100	92	940	310
4	MoCl ₅ -Ph ₃ Sb	91	78	690	220
5	MoCl ₅ -Ph ₃ Bi	100	87	640	230
6	MoCl₅−n-BuLi	31	29	650	330
7	MoCl ₅ -Et ₃ Al	25	2		
8	$Mo(CO)_6$ - CCl_4 - $h\nu^d$	80	72	1800	490

^aPolymerized in toluene at 30 °C for 24 h; [M]₀ = 0.50 M, [Cat] = [Cocat] = 20 mM. ^bPolymer: methanol-insoluble product. ^cDetermined by GPC. ^dPolymerized in CCl₄; [Mo(CO)₆] = 10 mM

Table V
Attempted Polymerization of 1-Chloro-3-propyl-1-hexyne $[ClC = CCH(n - C_3H_7)_2]^{a,b}$

catalyst	temp, °C	monomer convn, %	polymer yield, %°
MoCl ₅ -n-Bu ₄ Sn	30	42	12^d
MoCl₅−n-Bu₄Sn	60	77	3
$MoCl_5-n-Bu_4Sn$	80	86	0
$Mo(CO)_6$ - CCl_4 - $h\nu^e$	30	0	0
WCl_6-n -Bu ₄ Sn	30	3	0
WCl_6-n -Bu ₄ Sn	60	19	0
$W(CO)_6 - CCl_4 - h\nu^e$	30	0	0
NbCl ₅ -n-Bu ₄ Sn	80	16	0
TaCl ₅ -n-Bu ₄ Sn	80	6	0

^a Polymerized in toluene for 24 h; $[M]_0 = 0.50$ M, $[Cat] = [n-Bu_4Sn] = 20$ mM. ^bThe main product: linear oligomers.

^c Polymer: methanol-insoluble product. ${}^d\bar{M}_{\rm w} = 5000$ (GPC).

^e Polymerized in CCl_4 ; $[M(CO)_6] = 10$ mM.

polymerization behavior of this monomer is very similar to that of 1-chloro-1-alkynes with a shorter alkyl group such as 1-chloro-1-octyne. The appearance and mechanical properties of the polymers, however, changed appreciably with the length of the alkyl group (see below).

Polymerization of 1-Chloro-3-propyl-1-hexyne [ClC=CCH(n-C₃H₇)₂]. Polymerization of 1-chloro-3-propyl-1-hexyne, which possesses a sec-alkyl group, was attempted (Table V). Its polymer, however, could hardly be obtained with any catalysts. The yield of the polymer (methanol-insoluble product) produced with $MoCl_5-n-Bu_4Sn$ at 30 °C was only $\sim 10\%$, and the \overline{M}_w was only 5 \times 10³. Though the monomer conversion was increased by increasing temperature, no polymer formed at all at high temperatures around 80 °C. All of W, Nb, and Ta catalysts failed to produce a polymer.

1-Chloro-2-phenylacetylene polymerizes in high yields with Mo catalysts.^{5a,b} The very low polymerizability of 1-chloro-3-propyl-1-hexyne seems to be due to the steric effect of the sec-alkyl group being even larger than that of the phenyl group. Further, the fact that none of the 1-halo-1-alkynes including the present monomer gives any polymers with Nb or Ta catalysts suggests that these catalysts do not induce the polymerization of 1-halo-1-alkynes which has an electron-withdrawing halogen adjacent to the C≡C bond.

Copolymerizations of 1-Chloro-1-decyne with 2-Decyne and 1-Chloro-2-phenylacetylene. In order to clarify the effect of chlorine on monomer reactivity, copolymerizations of 1-chloro-1-decyne with 2-decyne and 1-chloro-2-phenylacetylene by MoCl_5 -n-Bu₄Sn were carried out. As is clear from Figure 4, the reactivity of 1-chloro-1-decyne in copolymerization is close to that of 2-decyne and is appreciably higher than that of 1-chloro-

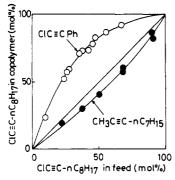


Figure 4. Composition curves for the copolymerizations of 1-chloro-1-decyne with 2-decyne and 1-chloro-2-phenylacetylene by $MoCl_5-n-Bu_4Sn$ (1:1) (in toluene, 30 °C, $[M]_{0,total} = 0.50$ M, $[MoCl_5] = 5-10$ mM).

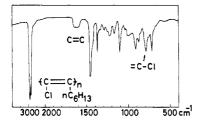


Figure 5. IR spectrum of poly(1-chloro-1-octyne) (KBr pellet, sample from Table I, no. 5).

Table VI Elemental Analysis Data of Poly(1-chloro-1-alkynes)^a

~(C(Cl)=				
$C(R)_{n}$	n-C ₄ H ₉	n-C ₆ H ₁₃	n-C ₈ H ₁₇	n-C ₁₄ H ₂₉
C, %	61.43	66.40	69.63	74.17
·	(61.81)	(66.43)	(69.55)	(74.82)
H, %	7.75	9.23	10.16	11.76
	(7.78)	(9.06)	(9.92)	(11.38)
Cl, %	30.02	24.34	20.48	13.63
-	(30.41)	(24.51)	(20.53)	(13.80)

^aSamples from Table I, no.s 2, 5, and 15, and Table IV, no. 2. Calculated values in parentheses.

2-phenylacetylene. The monomer reactivity ratios determined with the data in Figure 4 are as follows: $r_1=0.48\pm0.04$ and $r_2=1.12\pm0.12$ for the copolymerization of 1-chloro-1-decyne (M₁) with 2-decyne (M₂); $r_1=6.69\pm0.46$ and $r_2=0.52\pm0.06$ for the copolymerization of 1-chloro-1-decyne (M₁) with 1-chloro-2-phenylacetylene (M₂).

A general trend has been observed in copolymerization that, as an acetylene is sterically more crowded, it shows a lower relative reactivity.⁶ It is reasonable to assume that the steric effects of 1-chloro-1-decyne and 2-decyne are similar to each other and smaller than that of 1-chloro-2-phenylacetylene. Then the present result can also be explained in terms of the steric factor of monomers, and it can be said that neither the electronic effect of chlorine nor the resonance effect of phenyl group plays an important role to the reactivity in copolymerization.

In general, copolymerization helps modify polymer properties. It is known that poly(1-chloro-1-decyne) is soft and ductile, whereas poly(1-chloro-2-phenylacetylene) is hard and brittle. The When the copolymer composition was selected, mechanical properties of poly(1-chloro-1-decyne-co-1-chloro-2-phenylacetylene) could be arbitrarily controlled between those of the two homopolymers.

Polymer Structure. Values for the elemental analysis of poly(1-chloro-1-alkynes) were in satisfactory agreement with their theoretical ones (Table VI).

Figure 5 is the IR spectrum of poly(1-chloro-1-octyne). An absorption band due to C—C stretching is observed

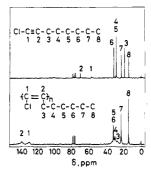


Figure 6. ¹³C NMR spectrum of the monomer and polymer of 1-chloro-1-octyne (CDCl₃ solution, polymer sample from Table I, no. 4).

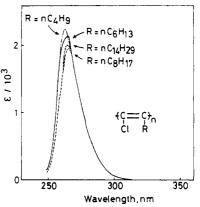


Figure 7. UV-visible spectra of poly(1-chloro-1-alkynes) (CCl_4 solution; samples from Table I, no.s 2, 5, and 15, and Table IV, no. 2).

at 1660–1600 cm⁻¹. The band at 800 cm⁻¹ is probably due to C-Cl stretching. Strong C-H stretching bands are seen at 2950 and 2850 cm⁻¹, while C-H deformation bands are at 1460 and 1380 cm⁻¹. The spectra of other poly(1-chloro-1-alkynes) were analogous.

In the 13 C NMR spectrum of the 1-chloro-1-octyne monomer are observed two acetylenic carbons at δ 69.8 and 57.2 (Figure 6). In the spectrum of its polymer, these peaks disappear, and instead two olefinic carbons appear at δ 136.8 and 129.0. The peaks of alkyl carbons in the side chain are seen in δ 35–15, an ordinary alkyl region. No other unexpected signals are observed. Other poly(1-chloro-1-alkynes) showed similar 13 C NMR spectra.

These IR and ¹³C NMR data are compatible with the idea that the present polymers have the alternating double bond structure as shown in Figures 5 and 6. No information on the geometric structure of the main chain, however, could be obtained from these spectra.

As seen in Figure 7, the UV-visible spectra of poly(1-chloro-1-alkynes) show a small absorption maximum ($\epsilon \sim 2000~{\rm M}^{-1}~{\rm cm}^{-1}$) around 280 nm and no absorption above 310 nm. These spectra correspond to the colorlessness of the polymers, indicating that alternating double bonds can conjugate only in a limited range owing to twisted main-chain conformations. The yellow color of poly(1-chloro-2-phenylacetylene)^{5a} should result from conjugation of the main chain with phenyl group.

A logarithmic plot of intrinsic viscosities ([η]) vs. $\bar{M}_{\rm w}$'s (data in Tables I and II) of poly(1-chloro-1-octyne) gave a good linear relationship. The least-squares treatment of the plot led to the equation

$$[\eta] = K\bar{M}_{\rm w}^{a}$$
 $(a = 0.923, K = 1.11 \times 10^{-5} \, \rm dL \cdot g^{-1})$

Exponent a in this equation is similar to those of other substituted polyacetylenes $(a = 0.9-1.1)^{2a}$ and larger than

spin den-

sity, g⁻¹

Table VII Several Properties of Poly(1-chloro-1-alkynes)^a

-(C(Cl)=	R					
$C(R))_n$	n-C ₄ H ₉	n-C ₆ H ₁₃	n-C ₈ H ₁₇	n-C ₁₄ H ₂₉		
	Heat Treat	ment (In Air, 1	20 °C, 20 h)			
$\bar{M}_{\rm w}/10^3$, ,			
before	890	870	940	630		
after	730	680	650	ь		
$\bar{M}_{\mathrm{n}}/10^3$						
before	590	510	550	320		
after	340	360	250	Ь		
$M_{\rm w}/M_{\rm n}$						
before	1.5	1.7	1.7	2.0		
after	2.1	1.9	2.6			
	Mechanica	l and Thermal	Properties			
E,c MPa	800	430	70			
$\gamma_{\mathrm{B}}^{},d}$ %	7	240	440			
$T_{\mathbf{z}}, e \circ \mathbb{C}$	190	150	90	-25		
softening	~310	~ 245	\sim 225	~ 150		
pt, °C						
	Electric a	and Magnetic F	Properties			
σ_{s}^{f} S·cm ⁻¹	2×10^{-17}	2×10^{-17}				

 a Samples from Table I, no.s 2, 5, and 15, and Table IV, no. 2. b The sample became partly insoluble after heat treatment. cE : Young's modulus (25 °C, data from ref 7). $^d\gamma_B$: elongation at break (25 °C, data from ref 7). eT_g : glass transition temperature (data from ref 7). $^f\sigma$: electrical conductivity (25 °C). g Unpaired-electron density (25 °C).

 $<1 \times 10^{15}$

those of usual vinyl polymers (a = 0.5-0.8). Consequently, it proves that this polymer takes a fairly expanded conformation in solution.

Polymer Properties. Properties of the four kinds of polymers from 1-chloro-1-hexyne, -1-octyne, -1-decyne, and -1-hexadecyne are described (samples from Table I, no. 2, 5, and 15, and Table IV, no. 2). The present polymers except poly(1-chloro-1-hexadecyne) had the form of white solid. Poly(1-chloro-1-hexadecyne) was a white rubbery material.

In general, these four polymers completely dissolved in hydrocarbons and halogenated hydrocarbons; their solvents are benzene, toluene, hexane, cyclohexane, CCl₄, CHCl₃, (CH₂Cl)₂, tetrahydrofuran, diethyl ether, etc., while their nonsolvents include ethyl acetate, methyl benzoate, acetone, acetophenone, acetonitrile, N,N-dimethylformamide, dimethyl sulfoxide, acetic acid, and ethanol.

Molecular weight changes by the heat treatment of poly(1-chloro-1-alkynes) in air at 120 °C for 20 h are shown in Table VII. The polymers of 1-chloro-1-hexyne, -1-octyne, and -1-decyne underwent only slight decreases in $\overline{M}_{\rm w}$ and $\overline{M}_{\rm n}$. They did not show any molecular weight change even when left in air at room temperature for 1 month. A major fraction of poly(1-chloro-1-hexadecyne) became insoluble in toluene after the heat treatment. Since this polymer has a long alkyl group and in turn a low glass transition temperature ($T_{\rm g}$; see below), it is inferred that a small number of free radicals thermally formed react with alkyl chains of other molecules, leading to cross-linking.

As shown in Figure 8, the present polymers did not lose weight below 250 °C in the thermogravimetric analysis (TGA) in air. Interestingly, almost no weight loss was observed below 400 °C in poly(1-chloro-1-hexadecyne). This seems to be because, owing to the presence of the long alkyl chain, radicals thermally formed effect cross-linking which prevents the weight loss of the polymer.

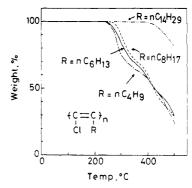


Figure 8. Thermogravimetric analysis of poly(1-chloro-1-alkynes) (in air, heating rate 10 °C/min; samples from Table I, no.s 2, 5, and 15, and Table IV, no. 2).

alkyl)]_n-<-[CCH₃-C(n-alkyl)]_n-<-[CCl-C(n-alkyl)]_n-<-[CCl-CPh]_n-. Poly(1-alkynes) are easily oxidized in air and suffer significant molecular weight decrease even at room temperature; their weight loss in TGA starts at as low a temperature as 150 °C. Hence, it is evident that introduction of a chlorine atom at the C₁ position of 1-alkynes greatly enhances the polymer stability . Poly(2-alkynes) gradually degrade in air at room temperature. Thus, it is noteworthy that poly(1-chloro-1-alkynes) are clearly more stable than poly(2-alkynes) and somewhat less stable than poly(1-chloro-2-phenylacetylene) which is one of the most stable polyacetylenes.

The wide-angle X-ray diffraction diagram (Cu K_{α} radiation) of poly(1-chloro-1-octyne) consisted of one broad peak: diffraction angle (2 θ) 23.6°, half-height width ($\Delta 2\theta$) 6.49°, $\Delta 2\theta/2\theta$ 0.275. Hence, it is concluded that poly(1-chloro-1-octyne) and probably other poly(1-chloro-1-alkynes) as well are amorphous.

Colorless, transparent, free-standing films are accessible from the polymers of 1-chloro-1-hexyne, -1-octyne, and -1-decyne by casting their toluene solutions on a glass plate. Poly(1-chloro-1-hexadecyne) film was too soft and ductile at room temperature to retain its form.

Mechanical properties of these polymers remarkably change with alkyl chain length: Poly(1-chloro-1-hexyne) requires much force to elongate [has a high Young's modulus (E)] and breaks at a slight elongation [shows a small elongation at break (γ_B)] (see Table VII); i.e., it is hard and brittle. In contrast, poly(1-chloro-1-decyne), whose alkyl chain length is twice that of poly(1-chloro-1-hexyne), has a much lower Young's modulus and can be elongated by as much as 440%; i.e., it is soft and ductile. Poly(1-chloro-1-hexadecyne) is too soft and ductile at room temperature to measure its tensile properties.

The T_g 's of the polymers of 1-chloro-1-hexyne, -1-octyne, and -1-decyne are 90 °C or above, and hence they are in the glass state at room temperature^{7a} (Table VII). In general, polyacetylenes have the stiff main chain, and therefore their T_g 's are high (usually ~ 200 °C).^{2a} In contrast, the T_g of poly(1-chloro-1-hexadecyne) is below room temperature. To our knowledge, this polymer is the first example of polyacetylenes whose T_g is lower than room temperature. The softening point of poly(1-chloro-1-alkynes) decreased sharply with an increase of alkyl group length.

The electrical conductivities (σ) of the polymers of 1-chloro-1-hexyne, -1-octyne, and -1-decyne at 25 °C were 10^{-17} – 10^{-16} S·cm⁻¹, lying in the insulator range ($<10^{-9}$ S·cm⁻¹). The unpaired-electron density of poly(1-chloro-1-octyne) at 25 °C was lower than the usual detection limit of 1×10^{15} spin·g⁻¹ in the electron spin resonance (ESR) measurement. These electric and magnetic properties of

Table VIII
Synthesis and Properties of 1-Chloro-Substituted
Acetylenes

R of ClC≡CR	yield, %	purity,4 %	bp, °C (mmHg)	d_4^{20} , g/mL
n-C ₄ H ₉ ^b	60	99	74 (180)	0.923
$n - C_6 H_{13}^{c}$	56	>99	52 (12)	0.908
$n-C_8H_{17}$	65	>99	83 (8)	0.890
$n-C_{14}H_{29}$	53	99	110 (2)	0.869
$CH(n-C_3H_7)_2$	46	97	60 (15)	0.810
Ph^{d}	75	>99	52 (3)	1.124

°By GC. ^bLit. bp 65–67 °C (125 mmHg). ⁸ °Lit. bp 61–62 °C (17 mmHg), d_4^{20} 0.912 g/mL. ⁹ ^dLit. bp 71 °C (16 mmHg), d_4^{18} 1.126 g/mL. ⁹

the present polymers, along with their color, indicate that their main chain is considerably twisted.

Conclusions

High molecular weight polymers were obtained for the first time from aliphatic linear 1-chloro-1-alkynes by use of Mo-based catalysts. The product polymers are white, totally soluble in many organic solvents, film-forming, and thermally fairly stable. Their mechanical properties can be easily controlled by the alkyl chain length. Unique functions (e.g., membrane separation) of these polymers based on their special properties are now under investigation.

Experimental Section

Monomer Synthesis. 1-Chloro-1-alkynes, including 1chloro-2-phenylacetylene, were prepared with reference to Brown's method.⁸ As an example, the procedure for synthesizing 1chloro-1-octyne is described in detail: The reaction was carried out in a 1-L three-necked flask fitted with a motor stirrer, reflax condenser, nitrogen inlet, and pressure-equalized dropping funnel. After flushing the system with dry nitrogen, the flask was charged with 1-octyne (0.40 mol, 44 g, 59 mL) and tetrahydrofuran (THF, 150 mL). After the flask had been cooled -50 °C, a hexane solution of n-butyllithium (1.6 M, 0.40 mol, 250 mL) was added dropwise from the dropping funnel, and stirring was continued further for 1 h. Then a THF solution of p-toluenesulfonyl chloride (0.40 mol, 76 g in 100 mL of THF) was added dropwise while the reaction mixture was kept at -50 °C. The reaction was completed by stirring at 0 °C for 1 h, which was confirmed by gas chromatography (GC). Water (10 mL) was carefully added, the reaction mixture was washed with water, and the water phase was extracted with hexane. The combined organic phase was dried over anhydrous sodium sulfate overnight, low volatiles were evaporated, and the product was distilled twice from calcium hydride.

Yields and properties of 1-chloro-1-alkyne monomers are shown in Table VIII. 1-Chloro-1-decyne, -1-hexadecyne, and -3-propyl-1-hexyne are new compounds. The other 1-chloro-1-alkynes in Table VIII had been prepared by various methods;³ the present process, however, seems the most convenient to carry out on a laboratory scale. Though molecular chlorine can be used to chlorinate 1-alkynyllithiums;¹⁰ use of p-toluenesulfonyl chloride is by far easier and safer. 3-Propyl-1-hexyne, the starting compound for the synthesis of 1-chloro-3-propyl-1-hexyne, was prepared by the literature method¹¹ (bp 58 °C/50 mmHg, yield 55%).

Catalyst Preparation and Polymerization. Transition metal compounds and organometallic cocatalysts were commercially obtained and used without further purification. Toluene as polymerization solvent was washed with dilute sulfuric acid, aqueous sodium hydroxide solution, and water, dried over calcium chloride overnight, and distilled twice from calcium hydride. Other polymerization solvents were purified similarly.

In general, catalyst preparation and polymerization were carried out under dry nitrogen in baked flasks equipped with a three-way stopcock. Catalysts composed of a transition metal chloride and an organometallic cocatalyst were allowed to stand (aged) in solution at polymerization temperature for 15 min just before use. Metal carbonyl based catalysts were prepared by UV irradiation of a CCl₄ solution of a metal carbonyl with a 200-W high-pressure

Hg lamp from a distance of 5 cm at 30 °C for 1 h.

The polymerization procedure is explained by the following typical example (see Table I, no. 5, for the results): In a 10-mL flask were placed 1-chloro-1-octyne (6.0 mmol, 0.87 g, 0.95 mL), dodecane (0.35 mL; as internal standard of GC), and toluene (4.7 mL). In a 30-mL flask, MoCl₅ (0.20 mmol, 55 mg) and n-Bu₄Sn (0.20 mmol, 69 mg, 66 μ L) were dissolved in toluene (5.0 mL), and the solution was left at 30 °C for 15 min. Then 5 mL of the monomer solution was added to the catalyst solution at 30 °C. After 24 h, polymerization was terminated by adding a mixture (3 mL) of toluene and methanol (volume ratio 4:1), and monomer conversion was determined by GC (silicone DC 3 m, 150 °C). The reaction mixture was diluted with toluene (20 mL) and poured into methanol (500 mL) under stirring. The precipitated polymer was filtered off and dried. Polymer yield was determined by gravimetry.

Polymer Characterization. $\bar{M}_{\rm w}$ and $\bar{M}_{\rm n}$ values of polymers were tentatively determined by GPC. GPC curves were observed with a Jasco Trirotar chromatograph (eluent, CHCl₃; columns, Shodex A802, A804, A806, and A807 polystyrene gels; polystyrene calibration). [η] values of polymers were measured in toluene at 30 °C. Elemental analysis was carried out at the Laboratory for Organic Elemental Microanalysis, Kyoto University. IR and ¹³C NMR spectra were recorded on a Shimadzu IR435 spectrophotometer and a Jeol FX90Q spectrometer, respectively. Thermogravimetric analysis was conducted with a Shimadzu DT20B thermal analyzer.

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Registry No. HC≡CC₈H₁₇, 100858-78-4; H₃CC≡CC₇H₁₆, 2384-70-5; ClC≡CPh, 1483-82-5; C₄H₉C≡CCl (homopolymer), 100858-76-2; C₆H₁₃C≡CCl (homopolymer), 100858-77-3; C₈H₁₇C≡CCl (homopolymer), 100858-79-5; C₁₄H₂₉C≡CCl (homopolymer), 108711-62-2; (C₃H₇)₂CHC≡CCl (homopolymer), 108711-64-4; ClC≡CPh (homopolymer), 81953-16-4; MoCl₅, 10241-05-1; Mo(CO)₆, 13939-06-5; CCl₄, 56-23-5; Bu₄Sn, 1461-25-2; Et₃SiH, 617-86-7; Ph₃Sh, 603-36-1; Ph₃Bi, 603-33-8; BuLi, 109-72-8; Et₃Al, 97-93-8; Ph₄Sn, 595-90-4; Ph₃SiH, 789-25-3; t-BuLi, 594-19-4

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